# Crystal Structure of Barium Titanyl Oxalate $\mathrm{BaTiO}\left(\mathrm{C}_{2} \mathrm{O}_{4}\right)_{2} \cdot \mathbf{4 . 5 \mathrm { H } _ { 2 } \mathrm { O }}$ 

M. LOUËR,* D. LOUËR,* F. J. GOTOR, $\dagger$ AND J. M. CRIADO $\dagger$<br>*Laboratoire de Cristallochimie (URA CNRS 254), Université de Rennes I, Avenue du Général Leclerc, 35042 Rennes cedex, France; and $\dagger$ Instituto Ciencias de Materiales (CSIC), P.B. 1065 Sevilla, Spain

Received November 12, 1990; in revised form February 13, 1991


#### Abstract

The hydrated phase of barium titanyl oxalate, $\mathrm{BaTiO}\left(\mathrm{C}_{2} \mathrm{O}_{4}\right)_{2} \cdot 4.5 \mathrm{H}_{2} \mathrm{O}$, is monoclinic with $a=13.382(2)$, $b=13.812(2), c=14.044(2) \AA, \beta=91.48(1)^{\circ}$, space group $P 2_{1} / n, Z=8$. The structure has been determined from Patterson and Fourier syntheses. The refinement was done by the method of leastsquares. The final $R$ values were $R=0.056$ and $R_{w}-0.079$ for 3261 unique reflections. The structure is characterized by a body-centered arrangement of finite groups consisting of four vertex-sharing $\mathrm{TiO}_{6}$ distorted octahedra. These groups are connected by barium polyhedra and oxalate groups. The study shows that the water moleculc number is 4.5 by formula rather than 4 reported previously. This conclusion is supported by TG measurements. © 1991 Academic Press, Inc.


## Introduction

The preparation of the hydrated phase of barium titanyl oxalate (BTO) for conversion by pyrolysis to stoichiometric barium titanate of high purity was described long ago (1). The composition reported for this material, in which the molar ratio of $\mathrm{TiO}_{2}$ to BaO is unity, is $\mathrm{BaTiO}\left(\mathrm{C}_{2} \mathrm{O}_{4}\right)_{2} \cdot 4 \mathrm{H}_{2} \mathrm{O}$. The scheme of its thermal decomposition has bcen investigated by several authors $(2,3)$. The stoichiometry and nature of this important precursor of barium titanate ceramics has been studied extensively and discussed in terms of factors influencing its synthesis (3). Although the powder diffraction pattern was described as "complicated" ( 3,4 ), a sample of BTO was recently investigated by means of high resolution conventional X-ray powder diffraction, using strictly monochromatic radiation (5). The indexing of the powder data was carried
out by means of the successive dichotomy method ( 6,7 ) and a monoclinic cell with high figures of merit $\left[M_{20}=46, F_{30}=107(0.0056\right.$, $50)$ ] was found. The large unit cell (2594.9 $\AA^{3}$ ) explains the high density of diffraction lines in the powder pattern and indexing confirms that BTO is a single phase. Moreover, in this work the average of the FWHM values of diffraction lines $\left(0.07^{\circ} 2 \theta\right)$, in the observed angular range, was comparable to the instrumental resolution function (8), which is an indication that structural imperfections are negligible and that coherently diffracting domains have large dimensions. We were motivated by these features to make a careful observation, by optical microscopy, of several samples of BTO, from which a small single crystal could be isolated. The present paper describes the determination of the crystal structure of the hydrated phase of barium titanyl oxalate and, as a consequence of the results obtained,
the initial stages of its thermal decomposition was again investigated and is also reported.

## Experimental Considerations

The double oxalate of barium and titanium was obtained by the method described in Ref. (1), using $\mathrm{BaCl}_{2} \cdot 2 \mathrm{H}_{2} \mathrm{O}, \mathrm{TiCl}_{4}$, and $\mathrm{H}_{2} \mathrm{C}_{2} \mathrm{O}_{4} \cdot 2 \mathrm{H}_{2} \mathrm{O}$ (Merck pro analysi) as starting materials. The TG analysis was carried out with a Rigaku Thermoflex TG-DSC. Powder samples of about 25 mg were spread cvenly, in a platinum sample holder. The samples were heated in air with a slow heating rate of $10^{\circ} \mathrm{C} \mathrm{hr}^{-1}$ in the range $18-800^{\circ} \mathrm{C}$.

## Structure Determination and Refinement

A small crystal ( $0.22 \times 0.24 \times 0.17 \mathrm{~mm})$ was selected from the powder and mounted on an Enraf-Nonius CAD4 diffractometer, with graphite-monochromated $\mathrm{Mo} K \alpha$ radiation. Systematic absences ( $h 0 l: h+l$ odd; $0 k 0: k$ odd) are consistent with the monoclinic $P 2_{1} / n$ space group. Cell constants, determined by the least-squares treatment of 25 reflections with $2 \theta$ values between 17.5 and $21.8^{\circ}$, are in agreement with the solution derived from the automatic indexing of the powder diffraction pattern (7). One set of intensities $(h: 0 \rightarrow 15, k: 0 \rightarrow 15, l: \overline{15} \rightarrow 15)$ was collected at room temperature by the $\theta-2 \theta$ scanning technique to give 3936 unique reflections with $\sin (\theta) / \lambda \leq 0.565 \AA^{-1}$. Within this set, 3261 reflections were considered as observed by means of the criterion $I>\sigma(I)$. The standard reflections ( 170 , 315,305 ) showed no significant variation during the data collection. Data were corrected for Lorentz and polarization effects. Due to the small size of the crystal no absorption correction was applied. Atomic scattering factors were taken from '"International Tables for X-ray Crystallography" (9). All calculations were performed with a MICROVAX 3100 computer by means of the SDP programs (10).

TABLE I
Crystallographic Data for BaTiO $\left(\mathrm{C}_{2} \mathrm{O}_{4}\right)_{2}$. $4.5 \mathrm{H}_{2} \mathrm{O}$

| Empirical formula | $\mathrm{Ba}_{1} \mathrm{Ti}_{1} \mathrm{C}_{4} \mathrm{O}_{13.5} \mathrm{H}_{9}$ |
| :---: | :---: |
| Formula weight | 458.27 |
| Crystal system | Monoclinic |
| Space group | $P 2_{1} / n$ |
| $a(\AA)$ | 13.382(2) |
| $b(\AA)$ | 13.812(2) |
| $c(\AA)$ | 14.044(2) |
| $\beta\left({ }^{\circ}\right.$ | 91.48(1) |
| $V\left(\AA^{3}\right)$ | 2594.9 |
| $Z$ | 8 |
| $D_{\text {calc }}\left(\mathrm{g} \cdot \mathrm{cm}^{-3}\right)$ | 2.34 |
| $D_{\text {obs }}\left(\mathrm{g} \cdot \mathrm{cm}^{-3}\right)[5]$ | 2.27 |
| $F_{000}$ | 1712 |
| Radiation ( $\AA$ ) | $\lambda=0.71069$ |
| Data collection | $h, k, \pm l$ |
| No. observations | 3261 |
| No. variables | 355 |
| $R$ | 0.056 |
| $R_{w}$ | 0.079 |

The structure was solved by the heavy atom method which yielded the positions of both independent Ba atoms. The positions of the remaining atoms were obtained from two successive Fourier maps. At this stage, the 37 independent nonhydrogen atoms correspond to the formula $\mathrm{BaTiO}\left(\mathrm{C}_{2} \mathrm{O}_{4}\right)_{2} \cdot 3.5$ $\mathrm{H}_{2} \mathrm{O}$. No subsequent peaks with suitable distances from Ba or Ti atoms were found. The $R$ and $R_{w}$ values ( $w^{-1}=\sigma^{2}(F)+$ ( $\left.0.04\left|F_{0}\right|\right)^{2}$ ), including anisotropic thermal parameters, were 6.9 and 10.2, respectively. The final Fourier-difference map showed two additional peaks with a height of 5.1 and $4.2 \mathrm{e} / \AA^{3}$. These peaks are not bonded to Ba and Ti atoms and could correspond to one additional water molecule, leading to the formula $\mathrm{BaTiO}\left(\mathrm{C}_{2} \mathrm{O}_{4}\right)_{2}$. $4.5 \mathrm{H}_{2} \mathrm{O}$. They were included in final cycles of the least-squares refinement and refined anisotropically. The $R$ and $R_{w}$ values then decreased to 5.6 and 7.9 , respectively. The refinement of occupation numbers for these two peaks gave the values 0.94 and 0.69 , but did not improve the $R$-factors, which

TABLE II
Positional and Thermal Parameters and Their Estimated Standard Deviations

| Atom | $x$ | $y$ | $z$ | $B_{\text {eq }}\left(\AA^{2}\right)$ |
| :---: | :---: | :---: | :---: | :---: |
| Ba1 | $0.53500(5)$ | 0.33772 (5) | 0.45467(4) | 1.87(1) |
| Ba 2 | $0.46724(5)$ | 0.06919(6) | $0.26211(5)$ | 2.62(1) |
| Til | 0.1277 (1) | 0.1226 (2) | 0.5274(1) | 1.72 (4) |
| Ti2 | 0.4182(1) | 0.3890 (1) | 0.8827(1) | 1.69 (4) |
| Cl | 0.0480(8) | 0.1828(8) | 0.7043(8) | 2.0(2) |
| C2 | 0.1548(8) | $0.1495(8)$ | 0.7319(8) | 2.0(2) |
| C3 | $0.3185(8)$ | 0.1763(9) | 0.4548(8) | 2.3(2) |
| C4 | 0.2754(9) | $0.2715(8)$ | 0.4887(8) | 2.6(2) |
| C5 | 0.4619(8) | 0.3158(8) | 0.6952(8) | 2.2(2) |
| C6 | $0.9696(8)$ | 0.0745 (8) | 0.1871 (7) | 1.9(2) |
| C7 | 0.2059(8) | 0.3477(8) | 0.8906(8) | 2.1(2) |
| C8 | 0.2592(8) | $0.2894(9)$ | 0.9717(8) | 2.2(2) |
| O1 | $0.4901(6)$ | $0.2886(6)$ | 0.2646(5) | 2.7(2) |
| O2 | 0.0269(5) | 0.1750(6) | $0.6159(5)$ | 2.2(2) |
| O3 | 0.2071 (5) | 0.1273 (6) | $0.6599(5)$ | 2.1(2) |
| O4 | $0.6831(6)$ | 0.3520 (6) | 0.3165 (5) | 2.7(2) |
| O5 | 0.4004(6) | $0.1733(6)$ | 0.4210 (6) | 2.8(2) |
| 06 | 0.2586 (5) | $0.1027(5)$ | 0.4650(5) | 2.1(2) |
| 07 | $0.3259(7)$ | 0.3453 (6) | $0.4821(7)$ | 3.6(2) |
| O8 | 0.1861 (5) | $0.2656(6)$ | 0.5182(5) | 2.4(2) |
| O9 | 0.4871 (7) | $0.2632(6)$ | 0.6292(5) | 3.7(2) |
| 010 | 0.4322(6) | 0.2864(6) | 0.7771 (5) | 2.3(2) |
| O11 | 0.4868(6) | $0.4651(6)$ | $0.6135(5)$ | 2.7(2) |
| 012 | $0.4511(6)$ | 0.4678(6) | 0.7667(5) | 2.2(2) |
| O13 | 0.2655(6) | 0.3905(6) | 0.8363(5) | 2.4(2) |
| 014 | $0.6147(6)$ | $0.1514(6)$ | $0.3866(6)$ | 2.8(2) |
| 015 | $0.3559(6)$ | 0.2941 (6) | $0.9685(5)$ | 2.4(2) |
| 016 | $0.7115(6)$ | 0.2562(6) | 0.5289(6) | 2.9(2) |
| 017 | $0.3937(5)$ | $0.4955(6)$ | $0.9524(5)$ | 2.1(2) |
| 018 | 0.0432(5) | 0.1399(6) | $0.4269(5)$ | $1.9(1)$ |
| Ow1 | 0.2548(9) | 0.085(1) | 0.2225(9) | 6.7(3) |
| Ow2 | $0.6474(7)$ | 0.1074(8) | $0.1704(7)$ | 4.3(2) |
| Ow3 | 0.4266 (2) | -0.092(1) | 0.150(1) | 12.1(6) |
| Ow4 | 0.5923(8) | -0.0716(8) | 0.320(1) | 8.0(4) |
| Ow5 | $0.439(1)$ | 0.1311 (9) | 0.0759(9) | 6.7(3) |
| Ow6 | 0.6993(9) | 0.4578(8) | $0.5234(9)$ | 5.9(3) |
| Ow7 | $0.388(1)$ | -0.065(1) | $0.385(1)$ | 11.0(4) |
| Ow8 | 0.325(1) | 0.428(1) | 0.172(1) | $8.5(4)$ |
| Ow9 | 0.366(2) | 0.076(2) | 0.908(1) | 11.8(6) |

Note. Anisotropically refined atoms are given in the form of the isotropic equivalent displacement parameter defined as: $\frac{4^{*}}{}\left[a^{2 *} B(1,1)\right.$ $+b^{2 *} B(2,2)+c^{2 *} B(3,3)+a b(\cos \gamma)^{*} B(1,2)+a c(\cos \beta)^{*} B(1,3)+$ $\left.b c(\cos \alpha)^{*} B(2,3)\right]$.

TABLE III
Bond Distances ( $\AA$ ) with Their Estimated Standard Deviations

| Bal | O1 | 2.806(4) | Ba 2 | Ow5 | 2.771(7) | C7 | C8 | 1.555(9) |
| :---: | :---: | :---: | :---: | :---: | :---: | :---: | :---: | :---: |
| Bal | O4 | $2.825(5)$ | Ba2 | Ow7 | 2.800 (8) | C1 | O1 ${ }^{\text {iii }}$ | 1.231(8) |
| Bal | O5 | 2.934(5) | Ti1 | O2 | 2.000 (4) | C1 | O2 | $1.272(8)$ |
| Ba1 | 07 | 2.842(5) | Ti1 | O3 | 2.119(4) | C2 | O3 | 1.285(7) |
| Bal | O9 | $2.755(5)$ | Ti1 | O6 | $2.003(4)$ | C2 | O4iii | 1.239(8) |
| Bal | O11 | $2.933(5)$ | Ti1 | 08 | $2.135(5)$ | C3 | 05 | $1.209(8)$ |
| Ba1 | O11 ${ }^{\text {iv }}$ | $2.905(5)$ | Ti1 | 017 ${ }^{\text {i }}$ | 1.808(5) | C3 | 06 | $1.308(8)$ |
| Bal | O14 | 2.962 (5) | Ti1 | 018 | $1.800(4)$ | C4 | O7 | 1.232(8) |
| Ba1 | 016 | $2.796(5)$ | Ti2 | 010 | $2.068(4)$ | C4 | O8 | 1.279(8) |
| Bal | Ow6 | 2.903(6) | Ti2 | 012 | 2.021 (4) | C5 | O9 | $1.233(8)$ |
| Ba 2 | O1 | 3.054(5) | Ti2 | 013 | 2.130 (4) | C5 | O10 | $1.295(8)$ |
| Ba 2 | O5 | 2.827(5) | Ti2 | 015 | $1.985(5)$ | C6 | 011 | 1.200 (8) |
| Ba2 | O14 | 2.840(5) | Ti2 | 017 | 1.807(4) | C6 | 012 | 1.294(8) |
| Ba 2 | Ow1 | 2.894(7) | Ti2 | O18 ${ }^{\text {ii }}$ | 1.814(4) | C7 | O13 | 1.268 (8) |
| Ba2 | Ow2 | 2.820(6) | C1 | C2 | $1.543(9)$ | C7 | O14 ${ }^{\text {iii }}$ | 1.222(8) |
| Ba2 | Ow3 | 2.788(9) | C3 | C4 | 1.52(1) | C8 | O15 | $1.300(8)$ |
| Ba2 | Ow4 | 2.681(6) | C5 | C6 | 1.528(9) | C8 | O16 ${ }^{\text {iii }}$ | 1.217(8) |

Note i: $1 / 2-x, y-1 / 2,1.5-z ;$ ii: $1 / 2+x, 1 / 2-y, 1 / 2+z$; iii: $x-1 / 2,1 / 2-y, 1 / 2+z ;$ iv: $1-x$, $1-y, 1-z$.
suggests that the chemical interpretation of this attempt is not obvious. Crystallographic data and final atomic parameters with equivalent temperature factors are given in Tables I and II, respectively. In Table II, O1-O16
correspond to the oxygen atoms of oxalate groups, O17 and O18 to the $\mathrm{O}^{2-}$ ions of TiO groups, Ow1-Ow7 to the water molecules and Ow8 and Ow9 to the additional nonbonded water molecules. Selected bond dis-

TABLE IV
Bond Angles $\left({ }^{\circ}\right)$ with Their Estimated Standard Deviations

| O2 | Til | O3 | 76.9(2) | O10 | Ti2 | 017 | 166.6(2) | C4 | C3 | O5 | 120.7(6) |
| :---: | :---: | :---: | :---: | :---: | :---: | :---: | :---: | :---: | :---: | :---: | :---: |
| O 2 | Til | O6 | 159.9(2) | O10 | Ti2 | O18 | 89.6(2) | C4 | C3 | 06 | 113.6(6) |
| O 2 | Til | 08 | 87.5(2) | 012 | Ti2 | 013 | 88.7(2) | O5 | C3 | O6 | 125.7(7) |
| O 2 | Til | 017 | 98.2(2) | 012 | Ti2 | O15 | 162.4(2) | C3 | C4 | 07 | 118.8(6) |
| O2 | Ti1 | 018 | 91.1(2) | 012 | Ti2 | O17 | 92.6(2) | C3 | C4 | O8 | 114.5(6) |
| O3 | Ti1 | O6 | 88.1(2) | 012 | Ti2 | 018 | 100.1(2) | O7 | C4 | O8 | 126.7(7) |
| O3 | Til | O8 | 81.3(2) | 013 | Ti2 | O15 | 77.0 (2) | C6 | C5 | 09 | 120.6(6) |
| O3 | Ti1 | 017 | 88.4(2) | 013 | Ti2 | 017 | 88.3(2) | C6 | C5 | 010 | 113.7(6) |
| 03 | Ti1 | 018 | 166.8(2) | 013 | Ti2 | O18 | 167.5(2) | O9 | C5 | 010 | 125.6(6) |
| O6 | Til | 08 | 76.9(2) | 015 | Ti2 | O17 | 97.1(2) | C5 | C6 | 011 | 122.3(6) |
| 06 | Ti1 | 017 | 94.6(2) | 015 | Ti2 | O18 | 92.6(2) | C5 | C6 | 012 | $111.8(5)$ |
| O6 | Til | 018 | 102.3(2) | 017 | Ti2 | O18 | 100.0(2) | Ol1 | C6 | 012 | 125.9(6) |
| O8 | Ti1 | 017 | 166.8(2) | C2 | C1 | O1 | 121.5(6) | C8 | C7 | 013 | 113.6 (5) |
| O8 | Ti1 | 018 | 93.0(2) | C2 | C1 | O2 | $113.5(5)$ | C8 | C7 | 014 | 118.4(6) |
| 017 | Til | 018 | 98.8(2) | O1 | C1 | O 2 | 125.0(6) | O 13 | C7 | 014 | 128.0(6) |
| 010 | Ti2 | 012 | 76.4(2) | C1 | C2 | O3 | 113.4(5) | C7 | C8 | 015 | 112.6(6) |
| 010 | Ti2 | 013 | 83.9(2) | C1 | C2 | O4 | 120.2(6) | C7 | C8 | 016 | 121.0(6) |
| O10 | Ti2 | 015 | 91.7(2) | O 3 | C2 | O4 | 126.5(6) | O15 | C8 | 016 | 126.4(6) |



Fig. 1. Stereoscopic view of the unit cell of BaTi $\mathrm{O}\left(\mathrm{C}_{2} \mathrm{O}_{4}\right)_{2} \cdot 4.5 \mathrm{H}_{2} \mathrm{O}$ along $c$, with $a$ horizontal and $b$ vertical.
tances and angles are listed in Tables III and IV.

## Results and Discussion

A stereoscopic view of the structure is shown in Fig. 1, from which its three-dimensional nature is readily apparent. Figure 2 shows different parts of the structure. Ba 1 (Fig. 2a) is bonded to 10 oxygen atoms arising from five $\mathrm{C}_{2} \mathrm{O}_{4}$ groups and one water molecule. The $\mathrm{Ba} 1-\mathrm{O}$ distances are in the range $2.75^{\circ}$ to $2.962 \AA$ (mean value: 2.866 $\AA$ ). Ba 2 (Fig. 2b) is bonded to 9 oxygen atoms due to three $\mathrm{C}_{2} \mathrm{O}_{4}$ groups and six water molecules with bond lengths ranging from 2.681 to $3.054 \AA$ (mean value: 2.830 $\AA$ ). The two Ti atoms are octahedrally bonded to two $\mathrm{O}^{2-}$ ions and two $\mathrm{C}_{2} \mathrm{O}_{4}$ groups (Fig. 2c). This coordination number of 6 for Ti with O or oxygen-containing ligands is the one most frequently encountered. The $\mathrm{Ti}-\mathrm{O}$ distances are in the range 1.800 to $2.135 \AA$. The values of the $\mathrm{Ba}-\mathrm{O}$ and $\mathrm{Ti}-\mathrm{O}$ distances and the characteristics of the $\mathrm{C}_{2} \mathrm{O}_{4}^{2-}$ groups are in good agreement with those reported in the literature (11-15). It can be noted that the distances $\mathrm{Ti}-\mathrm{O}^{2-}$ (mean value: $1.807(4) \AA$ ) are significantly shorter than the distances between Ti and the oxygen atoms of the oxalate groups (mean value: $2.058(4) \AA$ ). The four independent oxalate groups have the same coordination. Figure 2d shows the environment of
the oxalate group, which acts as a bidentate ligand to Ba and Ti and as a monodentate ligand to Ba 2 . Each water molecule, except Ow8 and Ow9, is bonded to one Ba atom. Although attempts to locate H atoms failed, no ambiguity exists for differentiating $\mathrm{O}^{2-}$ and water molecules. Moreover, according to Baur's criteria (16), the possible hydrogen bonds are listed in Table V , where $\mathrm{O}-\mathrm{O}$ distances less than $3.10 \AA$ are listed as well as the corresponding angles which are within acceptable limits when compared to the tetrahedral angles.

A way of looking at the crystal structure is to regard it as a body-centered arrangement of finite $\mathrm{Ti}_{4} \mathrm{O}_{20}$ groups consisting of four ver-tex-sharing $\mathrm{TiO}_{6}$ distorted octahedra (Fig. $2 \mathrm{e})$. The shared vertices are O 17 or O18. As shown in Fig. 1, these groups are connected by oxalate ions and Ba 2 polyhedra along the [100] and [010] directions and by oxalate ions and Ba 1 polyhedra along the [001] and [110] directions. This description, corresponding to the formula $\mathrm{BaTiO}\left(\mathrm{C}_{2} \mathrm{O}_{4}\right)_{2} \cdot 3.5 \mathrm{H}_{2} \mathrm{O}$, does not include the two last oxygen atoms (Ow8-Ow9 in Table II). These atoms, which correspond to one water molecule by chemical formula, are not bonded to the heavy atoms and are located in the vicinity of the Ba polyhedra. (The shortest distances to barium atoms are $\mathrm{Ba} 1-\mathrm{Ow} 9=4.66 \AA, \mathrm{Ba} 2-\mathrm{Ow} 8=$ $4.50 \AA$, and $\mathrm{Ba} 2-\mathrm{Ow} 9=3.88 \AA$.) Their distances from the nearest oxygen atoms and oxalate groups are consistent with hydrogen bonds (Table V). By including these oxygen atoms, structural analysis indicates that the composition of BTO has to be written BaTi $\mathrm{O}\left(\mathrm{C}_{2} \mathrm{O}_{4}\right)_{2} \cdot(3.5+1) \mathrm{H}_{2} \mathrm{O}$, while in previous studies, based on TG measurements, a tetrahydrate was generally reported $(2,3)$. In the TG analysis described by Gallagher and Schrey (2), the curve obtained at a heating rate of $60^{\circ} \mathrm{C} \mathrm{hr}{ }^{-1}$ revealed that the mass loss begins at the outset of the heating and no frank plateau is observed, making it difficult to evaluate the precise number of water molecules in the material. Due to the detection of


C



e


Fig. 2. The environment of Ba 1 (a), Ba 2 (b), Ti (c), and oxalate group (d); (e) represents the $\mathrm{Ti}_{4} \mathrm{O}_{20}$ group of four vertex-sharing octahedra.

TABLE V
Possible Hydrogen Bonds

| Distances $(\AA)$ |  | Angles $\left({ }^{\circ}\right)$ |  |
| :---: | :--- | :--- | ---: |
| Ow1-Ow6 | $2.893(7)$ | Ow6-Ow1-Ow8 | $104.7(3)$ |
| -Ow8 | $2.871(10)$ |  |  |
| Ow2-O8 | $2.821(6)$ | O8-Ow2-O13 | $121.0(3)$ |
| -O13 | $2.775(7)$ | O8-Ow2-Ow9 | $105.8(3)$ |
| -Ow9 | $2.762(12)$ | O13-Ow2 Ow9 | $112.1(3)$ |
| Ow3-Ow9 | $2.943(17)$ |  |  |
| Ow4-O9 | $2.955(8)$ | O3-Ow4-O9 | $94.7(2)$ |
| -O3 | $2.795(7)$ |  |  |
| Ow5-O2 | $2.775(7)$ | Ow1-Ow8-Ow7 | $74.9(3)$ |
| Ow6-O7 | $2.900(6)$ | Ow2-Ow9-Ow3 | $90.5(4)$ |
|  |  | Ow2-Ow9-Ow5 | $130.2(5)$ |
| Ow7-Ow8 | $2.867(13)$ | Ow2-Ow9-O4 | $95.6(3)$ |
|  |  | Ow3-Ow9-Ow5 | $83.4(5)$ |
| Ow9-Ow5 | $2.632(11)$ | Ow3-Ow9-O4 | $129.4(5)$ |
| -O4 | $2.910(11)$ | Ow5-Ow9-O4 | $125.8(4)$ |

two kinds of water molecule in the course of the structure determination, a TG analysis was carried out in air at a low heating rate ( $10^{\circ} \mathrm{C} \mathrm{hr}^{-1}$ ) to give a better resolution. The TG curve shown in Fig. 3 is similar to the result given by Gallagher and Schrey (2), except in the initial stages. Indeed, a slight inflection point is observed at $60^{\circ} \mathrm{C}$ for a value of $\Delta m / m_{0}$ close to $3.7 \%$, which is consistent with a loss of one water molecule ( $3.93 \%$ ). It could correspond to the loss of the water not


Fig. 3. Thermogravimetric curve (heating rate: $10^{\circ} \mathrm{C} \cdot \mathrm{h}^{-1}$ ).
coordinated to heavy atoms. The second inflection point, located approximately at $200^{\circ} \mathrm{C}$, corresponds to a mass loss of $17.40 \%$ which is in good agreement with the total loss of water. (Theoretical value for $4.5 \mathrm{H}_{2} \mathrm{O}$ content is $17.68 \%$.) In this second stage, the remaining 3.5 water molecules are eliminated from the precursor. Finally, the observed total mass loss is $49.00 \%$, which corresponds to the formation of barium titanate (theoretical value $49.10 \%$ ). The results of this TG study are in agreement with the chemical formula deduced from the structure analysis. Furthermore, samples of partially and completely dehydrated BTO , corresponding to $3.5,1.0,0.3$, and $0.0 \mathrm{II}_{2} \mathrm{O}$, respectively, were studied by powder diffraction. The results obtained show that the intensities of the diffraction lines decrease with decreasing water molecule content, yielding the anhydrous compound which is very poorly crystallized. This analysis suggests that no intermediate chemically defined phase exists and that the poorly crystalline feature of the anhydrous phase which is obtained can be due to the six water molecules removed from the polyhedra of the Ba 2 atom in the course of the thermal decomposition.

## References

1. W. S. Clabaugh, E. M. Swiggard, and R. Gilchrist, J. Res. Natl. Bur. Stand 56, 289 (1956).
2. P. K. Gallagher and F. Schrey, J. Amer. Ceram. Soc. 46, 567 (1963).
3. T. T. Fang and H. B. Lin, J. Amer. Ceram. Soc. 72, 1899 (1989).
4. K. Kudaka, K. Iizumi, and K. Sasaki, Amer. Ceram. Soc. Bull. 61, 1236 (1982).
5. D. Louër, A. Boultif, F. J. Gotor, and J. M. Criado, Powder Diffr. 5, 165 (1990).
6. D. LOUËR AND M. Louër, J. Applied Crystallogr. 5, 271 (1972).
7. A. Boultif and D. Louër. in "Collected Abstracts of Powder Diffraction Meeting," Toulouse, France, p. 135 (16-19 July, 1990).
8. D. LouËr and J. I. LangFord, J. Applied Crystallogr. 21, 430 (1988).
9. "International Tables for X-Ray Crystallography," Vol. IV, Kynoch Press, Birmingham (1974).
10. B. A. Frenz. "The Enraf-Nonius CAD-4 SDP."

Frenz and Associates, Inc., College Station, Texas, Enraf-Nonius, Delft, The Netherlands (1986).
11. Y. Dusausoy, J. Protas, J. C. Mutin, and G. Watelle, Acta Crystallogr. Sect. B 26, 1567 (1967).
12. J. Harada, T. Pedersen, and Z. Barnes, Acta Crystallogr. Sect. A 26, 236 (1970).
13. E. Tillmanns and W. H. Baur, Acta Crystallogr Sect. B 26, 1645 (1970).
14. E. Tillmanns, Acta Crystallogr Sect. B 30, 2894 (1974).
15. V. W. Hofmeister and E. Tillmanns, Acta Crystallogr. Sect. B 35, 1590 (1979).
16. W. H. Baur and A. A. Khan, Acta Crystallogr. Sect. B 26, 1584 (1970).

